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Dealloyed nanoporous platinum films: synthesis, characterization, and hydrogen sensing properties

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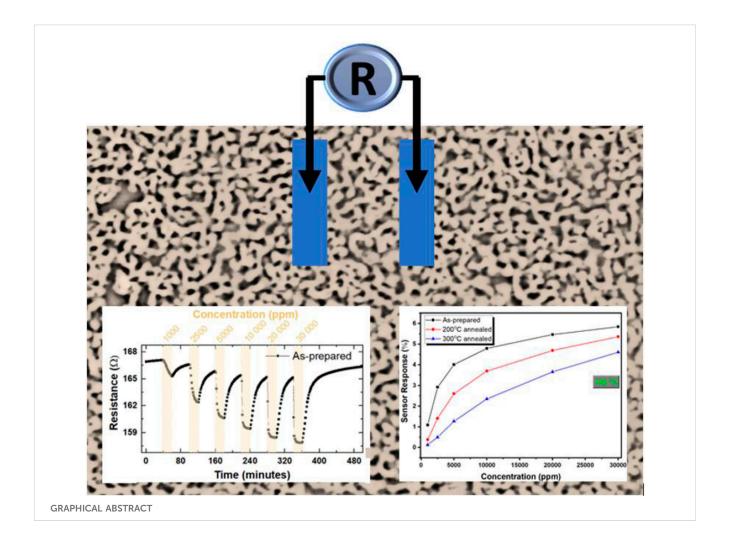
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Nanoporous platinum (Pt) films are synthesized at room temperature using a straightforward and cost-effective dealloying technique. This method is suitable for producing various nanoporous materials for diverse applications. Copper (Cu) atoms in the PtCu alloy films were selectively dissolved in a nitric acid solution, at varying times, to obtain nanoporous films. PtCu alloy thin films were then deposited onto a glass substrate utilizing the magnetron co-sputtering method with approximately 50 nm thickness. After 20 h dealloying in the acid solution, the residual Cu content in the alloy was less than 1% (atomic rate), and a regular nanoporous Pt structure was observed. The hydrogen detection properties of the nanoporous Pt films thus produced were investigated at various temperatures within a concentration range between 10 ppm and 5% hydrogen. The results demonstrated that a very high sensor response of 64 was obtained for the first exposure to 1% hydrogen at 150 °C, but the nanoporous Pt sensor resistance did not return to the baseline resistance. To utilize this nanoporous Pt film as a reversible hydrogen sensor, the film must be preexposed to hydrogen. After pre-exposure, the sensor response of the asprepared nanoporous Pt was approximately 4.5, resulting from exposure to 1% hydrogen at 150 °C, and the limit of detection was lower than 10 ppm. Data regarding the mechanism of the nanoporous Pt sensor device were clarified through surface scattering. The main contributions of this research are that sensing nanoporous film has a high surface-to-volume ratio, the sensor exhibited a very high initial response (~64) to 1% hydrogen at 150°C, the sensor mechanism is governed by surface scattering, and pre-exposure to hydrogen is needed for reversible sensing operation in practical usage.

KEYWORDS

nanoporous, platinum, dealloying method, co-sputtering, hydrogen sensor, resistive sensor



1 Introduction

Hydrogen is widely used in various industrial sectors, including semiconductor technologies, transportation, food, and chemistry (Abe et al., 2019; Ball et al., 2015). Recently, hydrogen fuel-cellpowered vehicles have been introduced by major automotive manufacturers such as BMW, Toyota, Honda, and Hyundai. However, it is crucial to take precautions against a small gas leak when using hydrogen, which is flammable at low temperatures, explosive when present in the air at a level of 4%, odorless, and invisible to the eye. Therefore, producing hydrogen sensors with a wide detection range, sensitive response, and easy installation is a prerequisite for minimizing the risks associated with working with hydrogen (Rane et al., 2015). Considering sensor performance (measuring concentration range, accuracy, response time, life time, and power consumption), metallic resistive hydrogen sensors are one of the best alternatives (Hubert et al., 2011; Sisman et al., 2024). Pt and Pd, which are more preferred as sensing materials in resistive hydrogen sensors, are also used as contact-doping elements and as decorative elements in/on sensitive layers in different types of sensors (Kilinc, 2013; Penner, 2017; Koo et al., 2020; Sahoo and Kale, 2021). Research on platinum (Pt), one of the sensing materials utilized in metallic resistive hydrogen sensors, has been comparatively limited compared to palladium (Pd)—

approximately 20 research articles focus on Pt-based resistive hydrogen sensors (Kilinc and Erkovan, 2023). Pt and Pt alloys have been used in resistive hydrogen sensor applications in nanostructure forms such as thin film (Cai et al., 2023; Patel et al., 1999; Tanaka et al., 2018; Kilinc et al., 2022; Erkovan et al., 2022; Sennik et al., 2016), bilayered films (Hassan et al., 2016), coreshell nanoparticle layers (Rajoua et al., 2018; Uddin et al., 2016), nanoporous film (Sener et al., 2023; Abburi and Yeh, 2012), and nanowires (Cao et al., 2019; Ding et al., 2017; Yoo et al., 2015; Yang et al., 2012). For example, Patel et al. (1999) investigated the hydrogen sensing properties of Pt films with 35 Å thickness. These films were prepared using electron beam evaporation on a SiO₂ substrate, depending on the temperature and hydrogen concentrations. An oxygen ratio of 5% in nitrogen was used as the carrier gas; the electrical resistance of the Pt film declined in the hydrogen environment, and the sensitivity of the Pt film was enhanced with temperature (Patel et al., 1999). Yang et al. (2012) produced Pd and Pt nanowires by the electrodeposition method and compared the hydrogen sensing of a single Pt nanowire with Pd nanowire and found a decline in the electrical resistance of the Pt nanowire when exposed to hydrogen. They determined a decrease in the electrical resistance of the Pt nanostructures in the hydrogen environment with the surface scattering phenomenon. Nevertheless, the increase in the electrical resistance of Pt nanostructures in the

hydrogen medium has been explained through grain boundary electron scattering (Rajoua et al., 2018), defect-dominated electron scattering (Cao et al., 2019), and PtH_x formation (Abburi and Yeh, 2012).

Nanoporous Pt structure films have been produced by coating Pt films on anodic aluminum oxide (AAO) nanotemplates and investigating their hydrogen sensing properties (Sener et al., 2023). The resistance of the nanoporous Pt film decreased in a hydrogen atmosphere, yielding approximately 13% sensitivity to 1% hydrogen at room temperature (Sener et al., 2023). Abburi and Yeh (2012) produced nanoporous Pt film structures from a PtCu alloy using dealloying and investigated their hydrogen sensing properties, which depend on pore size and temperature. They reported a sensitivity of 3.5% for a nanoporous Pt film with a pore size of 35 nm. Conversely, they attributed the observed increase in the resistance of the nanoporous Pt film in a hydrogen atmosphere to the formation of PtH_x. In this study, although nanoporous Pt films were fabricated using a similar method, a decrease in the resistance of the films was observed in the hydrogen environment. To the best of our knowledge, the highest sensitivity reported among Pt-based resistive hydrogen sensors was achieved upon the film's initial exposure to hydrogen. The hydrogen sensing properties of nanoporous Pt films were systematically investigated in terms of operating temperature, hydrogen concentration, and annealing temperature. Furthermore, the sensing mechanism of the nanoporous Pt film sensors was examined and discussed.

This study reports a highly sensitive nanoporous platinum (Pt) film hydrogen sensor prepared through a straightforward dealloying of magnetron co-sputtered PtCu thin films. The sensor exhibited a very high initial response (~64) to 1% hydrogen at 150 °C—the highest value ever reported for Pt-based resistive hydrogen sensors. In contrast to earlier reports, the sensor mechanism is governed by surface scattering, leading to a resistance decrease upon exposure to hydrogen. A new pre-hydrogenation process is also presented for a reversible sensing operation, which facilitates practical usability.

2 Materials and methods

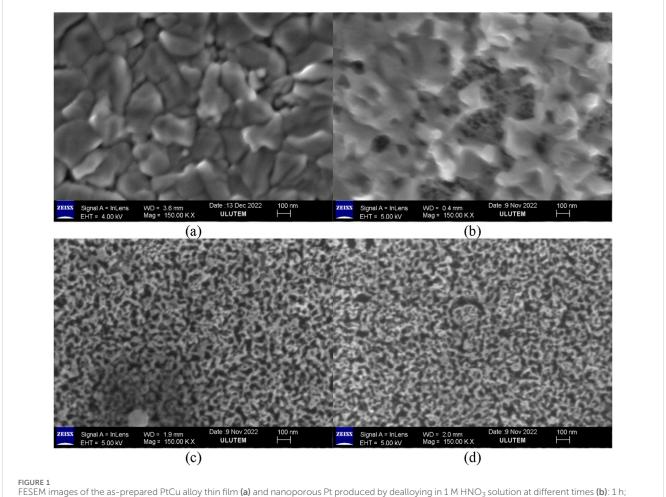
The preliminary results were presented at a conference (9th International Electronic Conference on Sensors and Applications) and published in its proceedings (Sener et al., 2022). Platinum-copper (PtCu) alloy films, approximately 50 nm thick, were coated onto a microscope glass slide using the magnetron cosputtering method with an atomic ratio of 15:85, respectively. All metallic film depositions were performed utilizing a NANOVAK 400 physical vapor deposition (PVD) system configured with dual sputtering and dual thermal evaporation sources. The glass was used as the substrate material, with a series of meticulous cleaning processes being employed prior to PtCu coating. The cleaning regimen began with subjecting the substrates to a sequence of solvents: acetone, isopropyl alcohol, and ultra-pure water (18 M Ω). Each solvent was applied for 10 min, 10 min, and 10 min, within an ultrasonic bath. Subsequent to this cleansing, the substrates were meticulously dried using a stream of nitrogen (N2) gas. In the final stage of the cleaning process, an ultrasonic plasma cleaner was engaged to effectively eliminate any residual contaminants. Magnetron sputtering was used to initiate the film growth process, forming a film with a thickness of 50 nm. After a base pressure of 1.6x10⁻⁶ Torr was obtained in the PVD chamber, approximately 3 sccm ultra-pure argon (Ar, purity: 99,999%) gas was injected into the chamber to maintain a growth pressure of 3 mTorr. Then, co-deposition was achieved using a radio frequency (RF) and direct current (DC) power level of 50 Watts with Cu and Pt targets, respectively. PtCu alloy films were dealloyed in 1 M nitric acid (HNO₃) solution for various times (from 15 min to 20 h) to obtain nanoporous Pt films. The fabrication of nanoporous Pt using dealloying is schematically illustrated in Supplementary Figure S1a. The nanoporous Pt films, which were dealloyed for 5 h, were annealed at temperatures of 200-300 °C to determine the annealing effect on structure and hydrogen sensing. Scanning electron microscopy (SEM, Zeiss LEO-EVO 40), field emission scanning electron microscopy (FESEM, Zeiss - Gemini 300), energy-dispersive X-ray spectroscopy (EDX) attached to the FESEM or SEM, X-ray photoelectron spectroscopy (XPS, Specs-Flex), and an X-ray diffraction (XRD, Rigaku - D/Max 2,200) spectrum were used to clarify the morphologies, crystal structures, composition of the nanoporous Pt and PtCu alloy films, and electronic structures.

Before determining the hydrogen sensing properties, two silver (Ag) contact electrodes were deposited onto the nanoporous Pt films via thermal evaporation with a shadow mask. Supplementary Figure S1b shows a schematic diagram of the measurement setup for a resistive gas sensor device. Continuous measurement of nanoporous Pt's electrical resistance was performed using a two-point probe configuration connected to a Keithley 2700 multimeter, while the atmosphere within a purpose-built measurement cell was systematically altered. The measurement cell consisted of a flowtype aluminum chamber that incorporated an integrated heatable sample holder. The precise regulation of hydrogen gas concentration, ranging from 10 ppm to 5%, was achieved by utilizing two mass flow controllers (Alicat). A Lakeshore 335 temperature controller facilitated the precise adjustment of temperature between 25 °C and 150 °C. All measurement data were logged using a LabVIEW-based data acquisition system connected to a personal computer via a general purpose interface bus (GPIB).

3 Results and discussion

3.1 Structural characterization

Figure 1 presents the FESEM images of the as-prepared PtCu alloy thin film and the nanoporous Pt films obtained by dealloying the alloy in a 1 M HNO₃ solution for durations of 1 to 20 h. Dealloying is a critical process in the fabrication of the nanoporous platinum (Pt) films herein. Following the deposition of PtCu alloy thin films via magnetron co-sputtering, dealloying was achieved by soaking the films in a 1 M nitric acid (HNO₃) solution for varying periods of time. This involved the selective dissolution of less noble copper (Cu) atoms in the alloy to create a three-dimensional, bicontinuous nanoporous Pt network. The resulting structure has a high surface-to-volume ratio and interconnected porosity at the nanoscale. These properties are essential for enhancing gas—solid interactions and achieving hydrogen adsorption. The dealloying



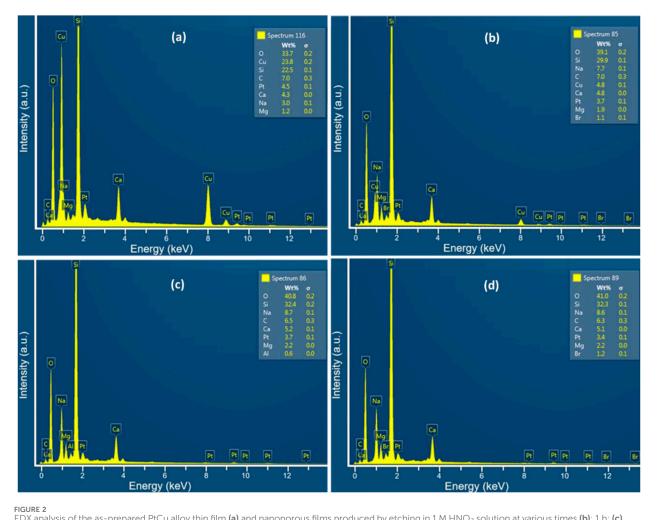
(c) 5 h; (d) 20 h.

process also provides an easy, room-temperature, scalable route to create functional nanostructured films without relying on templates or lithographic patterning, rendering it extremely viable for sensor applications. The as-prepared PtCu alloy films have a large-grained structure with little porosity (Figure 1a). During the dealloying of PtCu films with HNO₃ solution for 1, 5, and 20 h, nanoporous film was observed, and the porosity increased with increasing time (Figures 1b-d, respectively). A more detailed time-dependent surface examination was performed by SEM, and SEM images at 80KX magnification are provided in Supplementary Figure S2. Based on the SEM and FESEM results (Supplementary Figure S2b-d), it is observed that the amount of abrasion on the surface gradually increased when the dealloying time of the PtCu alloy film in HNO₃ solution was increased up to 2 h. On the other hand, when the PtCu alloy film was dealloyed in HNO₃ solution for 5 h or longer, the surface morphology of the nanopores appeared largely unchanged, with the porosity remaining approximately constant. This consistency is evident in Figures 1c, d and also in Supplementary Figures S2e, f.

The EDX results of the as-prepared PtCu alloy thin film (a) and nanoporous film produced by dealloying in 1 M HNO₃ solution at different times (b: 1 h, c: 2 h, d: 20 h) are presented in Figure 2. Peaks of Si, Ca, Na, Mg, Al, O, Cu, and Pt elements were detected in the

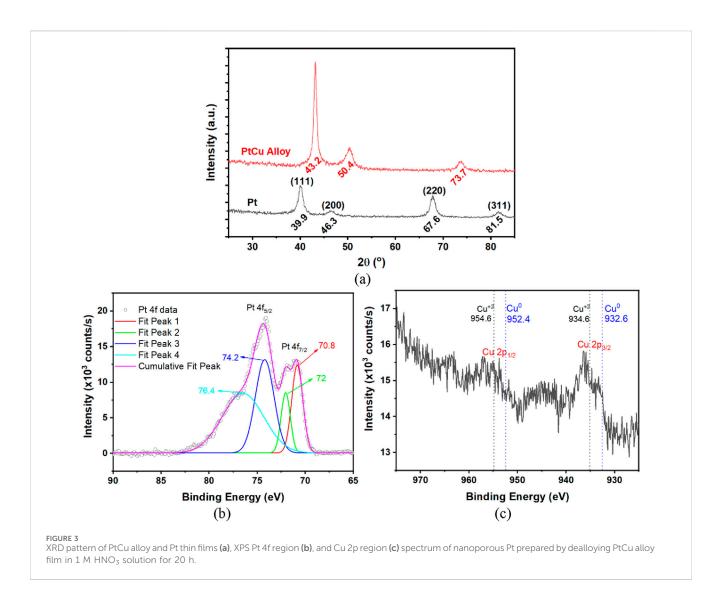
samples prepared on the glass substrate according to the EDX spectrums. The observed peaks corresponding to Ca, Na, Mg, Al, and Si elements were attributed to the inherent material composition of the microscopic glass substrate. Conversely, the Cu and Pt peaks originated from the deposited PtCu alloy coating. The atomic ratio of the PtCu alloy was calculated from the EDX spectrum by considering only the percentages of Cu and Pt and was noted to be 15.9:84.1 for the as-prepared PtCu alloy thin film. According to Figure 2b, the copper ratio in the alloy decreased to approximately 56% for the sample that was dealloyed in a HNO₃ solution for 1 h. In PtCu, alloys dealloyed in a 1 M HNO₃ solution for 5 and 20 h; there were no Cu peaks in the EDX spectrums (Figures 2c, d). This can be attributed to the possibility that either no Cu atoms remain in the structure or that the residual Cu concentration is below the detection limit of the EDX instrument. More detailed analyses conducted using X-ray photoelectron spectroscopy (XPS) support the latter explanation, indicating that Cu residues persist even in the film dealloyed for 20 h. Moreover, this observation supports the fact that the surface of nanoporous Pt, which had been dealloyed for 5 h or more, was the same in SEM and FESEM images.

The X-ray diffraction (XRD) graph of the PtCu alloy film and Pt thin film is given in Figure 3a. When the diffraction pattern was



EDX analysis of the as-prepared PtCu alloy thin film (a) and nanoporous films produced by etching in 1 M HNO₃ solution at various times (b): 1 h; (c) 5 h; (d) 20 h.

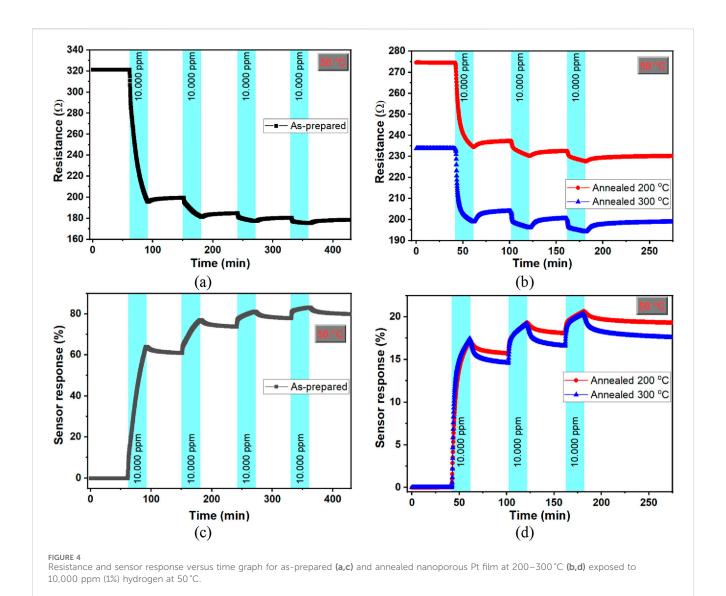
examined, it was determined that the diffraction peaks at 39.9°, 46.3°, 67.6°, and 81.5° of the Pt film belonged to the planes (111), (200), (220), and (311), respectively, and these peaks reveal Pt film in face-centered cubic (fcc) structure. As shown in Figure 3a, the diffraction peaks of the PtCu alloy exhibit a shift toward higher angles. This could be ascribed to the shrinkage in the lattice structure resulting from the replacement of small Cu atoms with larger Pt atoms (Kang et al., 2015). Similar shifts in diffraction peaks have been observed in previous studies of various PtCu alloy nanostructures, including nanoparticles, nanocrystals, nanoclusters (Jiang et al., 2013; Oezaslan et al., 2011; Toshima and Wang, 1994). The diffraction peak at 42.2° corresponded to the (111) lattice plane, which was reported for PtCu3 (1:3) alloy nanoparticles (Weihua et al., 2006). In our case, a diffraction peak at 43.2° corresponding to the (111) lattice plane for the PtCu alloy film was obtained. This indicates a Cu atom proportion of over 75% in the alloy. XPS survey spectra for nanoporous Pt fabricated by dealloying PtCu alloy film in a 1 M HNO3 solution for 20 h are shown in Supplementary Figure S3. We can clearly observe Cu, Pt, C, Si, and O peaks here. The XPS spectra in the Pt 4f and Cu 2p regions for the nanoporous Pt film that were observed by dealloying the PtCu alloy film in the 1 M HNO₃ solution for 20 h are presented in Figures 3b, c, respectively. The Pt region XPS spectra have two peaks at approximately 70.8 eV and 72 eV, corresponding to the Pt 4f7/2 core level of metallic Pt° and Pt2+ (PtO), respectively. Additionally, two peaks at approximately 74.2 eV and 76.4 eV correspond to the Pt 4f5/2 core level of Pto and Pt2+ (Figure 3b). The Cu XPS spectra show that the Cu 2p3/2 peak contains two peaks at 932.4 eV and 934.1 eV, corresponding to a small metallic Cu° and Cu²⁺ (CuO or Cu(OH)₂) (Figure 3c). Similar Cu⁺² and Pt⁺² oxide peaks were found for PtCu nanodendrites (Du et al., 2018), coreshell-structured nanoporous PtCu (Qiu et al., 2015b), nanoporous PtCu bimetallic microwire (Qiu et al., 2015a), and nanoporous Pt prepared by dealloying PtCu thin film (Giarratano et al., 2018). As a result, approximately 0.7% Cu remained in the Pt nanoporous films obtained by dealloying PtCu alloy in acid solution for 20 h. In PtCu alloys dealloyed in 1 M HNO3 solution for both 5 and 20 h, no Cu peaks were observed in the EDX spectra (Figures 2c,d). This indicates that the residual Cu content likely falls below the detection limit of the EDX technique. XPS analysis confirmed this, revealing that approximately 0.7% Cu was still present in the 20-h dealloyed film, thereby demonstrating the superior



sensitivity of XPS for trace element analysis. There are also small amounts of PtO and CuO oxide structures in the nanoporous film.

3.2 Hydrogen sensing

Nanoporous Pt film prepared by dealloying PtCu alloy film in the 1 M HNO₃ solution for 5 h was chosen to investigate hydrogengas-sensing properties and examine the annealing effect on the hydrogen-sensing properties. Nanoporous Pt samples were annealed at 200 °C and 300 °C. Figure 4a shows the resistance change of the as-prepared nanoporous Pt film during exposure to four times 10,000 ppm (1%) hydrogen and dry air cleaning at 50 °C. In the initial instance of hydrogen exposure to the nanoporous Pt sensor device, a decrease in resistance from 321 Ω to 196 Ω was observed. Then, although the measurement cell was purged with dry air for 2 h, the resistance increased slowly and did not recover to the baseline resistance. The resistance decreased when exposed to 10,000 ppm hydrogen for the second, third, and fourth times, but the resistance change was small compared to the first time. There was no recovery during the dry air cleaning. A similar behavior was obtained for Pt nanoporous structures annealed at 200 °C and 300 °C. Figure 4b presents the resistance versus time profile of the nanoporous Pt films annealed at 200 °C and 300 °C, subsequently exposed to dry air and 10,000 ppm hydrogen at 50 °C. In order to desorb hydrogen from the Pt nanoporous material, the devices were heated to 160 °C, kept at this temperature for 1 h, and then cooled to room temperature under a dry air flow. Even so, the resistance value did not recover to the first baseline value. This can be attributed, not to one reason but to a combination of the following reasons: the Cu/CuO remaining in the nanoporous Pt structure, the reduction of PtO in the nanoporous structure to metallic Pt, the hydrogens diffusing into the nanopores forming chemical bonds with the structure, and disordered nanoporous structure with factors such as defects, lattice expansioncontraction, and strain. Hakamada et al. (2011) prepared nanoporous Pt with a dealloying method and investigated high pressure hydrogen absorption/desorption properties. They found abnormal hydrogen absorption/desorption features and explained this by the relaxation/preservation of lattice strain in Pt hydride using first-principles calculations. In addition, Maya et al. (1999) prepared a few µm thick platinum dioxide (PtO₂) films using the RF magnetron sputtering method and investigated the reduction properties of these films under 4% hydrogen flow or the



electrochemical reduction features at room temperature. They observed the reduction of the PtO_2 film to a metallic Pt film through exposure to hydrogen or by electrochemical means. On the other hand, the baseline resistances of the as-prepared, 200 °C, and 300 °C annealed nanoporous Pt were 321 Ω , 274 Ω , and 233 Ω , respectively (Figures 4a,b). The decrease in the baseline resistance of nanoporous Pt with increasing annealing temperature could be explained by the structural reorganization of the nanoporous material. Figures 4c, d reveal the sensor response of as-prepared, 200 °C, and 300 °C annealed nanoporous Pt, which was calculated using Equation 1:

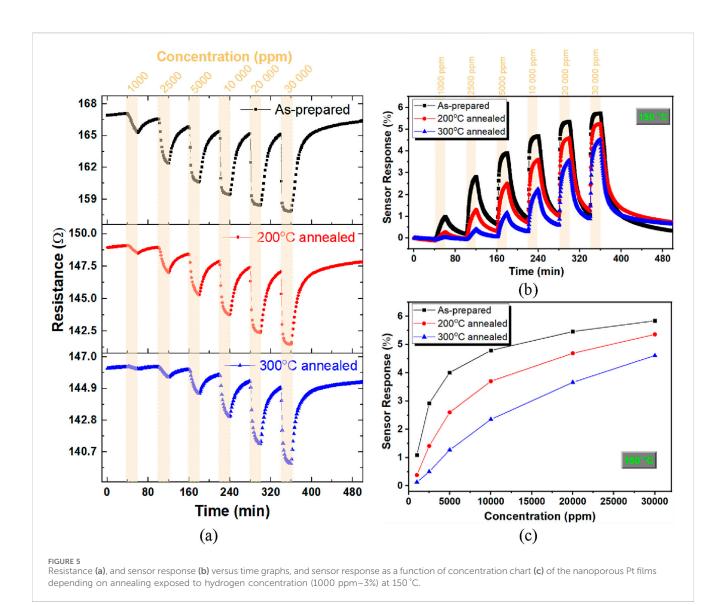
Sensor response (%) =
$$\frac{R_0 - R_H}{R_H} \times 100$$
 (1)

where R_0 is the resistance of the nanoporous device under dry air flow and R_H is the resistance of the nanoporous device exposed to the indicated hydrogen concentration. When the first 10,000 ppm of hydrogen was exposed to sensor devices, the sensor responses of the as-prepared and annealed nanoporous Pt were obtained approximately as 64% and 17%, respectively. The sensor

responses of nanoporous Pt annealed at 200 °C and 300 °C were almost the same. The decrease in the sensor response with annealing can be explained by the reorganization of the nanoporous structure during annealing, which decreases the number of sites where hydrogen can adsorb. The observed sensor response represents the highest value reported to date for Pt-based resistive hydrogen sensors, surpassing those listed in Kilinc et al. (2024), Sener et al. (2023), and Kilinc et al. (2022). Cao et al. (2019) obtained a sensor response of 18.4 during exposure to 1% hydrogen for a Pt nanowire array fabricated with a focused ion beam. Rajoua et al. (2018) prepared a Pt-Au core-shell nanoparticle layer using the Langmuir-Blodgett method and found the sensor response to be approximately 15 when exposed to 10% hydrogen. However, this nanoporous Pt sensor was not reversible and needed pre-exposure to hydrogen. Therefore, nanoporous Pt sensor devices were exposed to 4% hydrogen for 5 h at room temperature, and then the sensor devices showed a reversible response to hydrogen.

Figure 5a illustrates the variation in resistance of as-prepared, 200 °C, and 300 °C annealed nanoporous Pt films upon exposure to varying hydrogen concentrations ranging from 1,000 ppm to 3% at

10 3389/fnano 2025 1599264 Sener et al



an operating temperature of 150 °C. Upon exposure to 1,000 ppm hydrogen, a reduction in the electrical resistance of the nanoporous Pt films was observed. Subsequently, the resistance exhibited a slow, incremental increase during the purging process with high-purity dry air. These trends were consistently reproduced for the remaining hydrogen concentrations examined. Notably, an increase in hydrogen concentration resulted in a corresponding amplification of the resistance change observed in the nanoporous Pt films. The reactions occurring on the nanoporous Pt film surface in different atmospheres are as follows (Gland et al., 1982; Sachs et al., 2001):

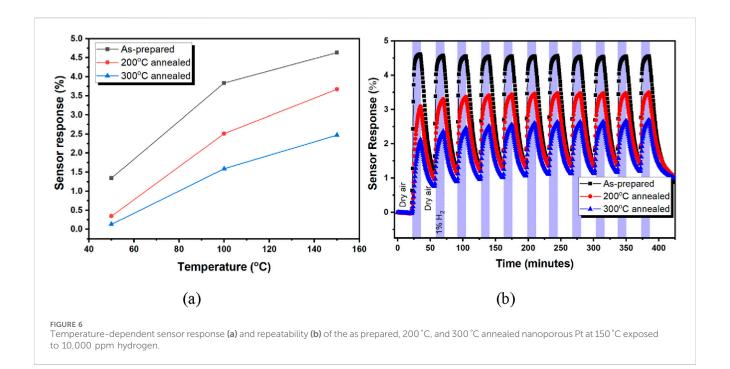
$$Pt + \frac{1}{2}O_2 \to Pt - O \tag{2}$$

$$Pt - O + \frac{3}{2}H_{2} \to Pt - H + H_{2}O$$

$$Pt - H + \frac{3}{2}O_{2} \to Pt - O + H_{2}O$$
(4)

$$Pt - H + \frac{3}{2}O_2 \rightarrow Pt - O + H_2O$$
 (4)

The proposed mechanism for hydrogen sensing is as follows. Under a dry air flow, the surface of the nanoporous Pt film becomes saturated with adsorbed oxygen species (chemical reaction in Equation 2), which influences its electrical resistance. Upon exposure of the nanoporous Pt film to hydrogen, a displacement reaction occurred in which hydrogen atoms interacted with and displaced adsorbed oxygen atoms (chemical reaction in Equation 3), leading to a reduction in electron surface scattering. Therefore, the decrease in the resistance of the nanoporous Pt film can be determined by the surface scattering phenomenon. During the recovery in dry air, oxygen molecules interacted with the adsorbed hydrogen, displacing hydrogen atoms and creating a reversible reaction cycle (chemical reaction in Equation 4). Parallel trends in hydrogen sensing have been noted concerning Pt nanowires (Yang et al., 2012; Yoo et al., 2015; Ding et al., 2017), nanoporous Pt on AAO nanotemplate (Sener et al., 2023), and Pt thin films (Sennik et al., 2016; Patel et al., 1999; Tanaka et al., 2018). In addition, the annealing effect on the baseline resistance and the sensor response of nanoporous Pt thin films are similar to the results before pre-exposure to 4% hydrogen, as mentioned in the previous paragraph. Figure 5b shows the sensor response versus time graph for the as-prepared, 200 °C, and 300°C annealed nanoporous Pt films exposed to various hydrogen concentrations at 150 °C. It has been established that the sensor responses of all nanoporous Pt films exhibited an increase in



response as the concentration was enhanced. The as-prepared nanoporous Pt film showed the highest sensor response in the measured concentration and temperature intervals. Figure 5c shows the sensor response as a function of the concentration curve for the asprepared, 200 °C, and 300 °C annealed nanoporous Pt at 150 °C. For example, a wide hydrogen concentration range from 10 ppm to 50,000 ppm tests was carried out for the as-prepared nanoporous Pt film as given in the preliminary results that presented at a conference and published in its proceedings (Sener et al., 2022). Therefore, the limit of detection for the as-prepared nanoporous Pt film was lower than 10 ppm.

Figure 6a presents the temperature-dependent sensor response for the as-prepared, 200 °C, and 300 °C annealed nanoporous Pt at 150 °C exposed to 10,000 ppm hydrogen. The sensor response of all nanoporous sensor devices increased with temperature. Figure 6b shows the repeatability of the as-prepared, 200 °C, and 300 °C annealed nanoporous Pt at 150 °C exposed to 11 times 10,000 ppm hydrogen. The sensor response variation was below 10% for 11 consecutive 10,000 ppm hydrogen injections for approximately 7 h, and the average sensor responses were obtained as 4.55, 3.39, and 2.51 for the as-prepared, 200 °C, and 300 °C annealed nanoporous Pt, respectively.

4 Conclusion

In the present study, a novel avenue for hydrogen sensing was unveiled through the fabrication of nanoporous platinum (Pt) films using the dealloying technique. Employing the magnetron cosputtering approach, platinum-copper (PtCu) alloy films approximately 50 nm thick were meticulously deposited onto glass substrates. The central step involved the strategic immersion of PtCu alloy films in a 1 M nitric acid solution for varying durations for the subsequent transformation into

nanoporous Pt films. Remarkably, after a 5-h and more immersion period, an intriguing outcome surfaced—the almost complete removal of Cu from the alloy matrix, resulting in a nanoporous Pt film distinguished by a well-defined and uniform pore structure. The primary objective of the study was to assess the performance of the nanoporous Pt film as a hydrogen sensor across a broad range of concentrations and temperatures. A key finding emerged as the film demonstrated an impressive sensor response of approximately 4.5 when subjected to 10,000 ppm hydrogen concentration. Furthermore, the underlying sensing mechanism of the nanoporous Pt film was attributed to the surface scattering phenomenon, whereby hydrogen interaction induces changes in electrical conductivity, leading to measurable alterations in resistance. The culmination of this research highlights the remarkable promise of nanoporous Pt films as a robust candidate for hydrogen sensing applications. Nanoporous Pt film sensors could potentially be used in leakage detection, breath hydrogen analysis, and transformer oil degradation.

Data availability statement

The raw data supporting the conclusions of this article will be made available by the authors without undue reservation.

Author contributions

MS: Data curation, Methodology, Writing – original draft, Investigation, Visualization, Validation, Writing – review and editing, Conceptualization, Formal Analysis. AA: Data curation, Writing – original draft, Investigation, Conceptualization, Writing – review and editing, Validation, Methodology, Formal Analysis, RZ: Formal Analysis, Conceptualization, Methodology, Funding

acquisition, Writing – review and editing, Supervision, Writing – original draft, Resources. NK: Formal Analysis, Writing – review and editing, Methodology, Validation, Project administration, Supervision, Conceptualization, Software, Funding acquisition, Writing – original draft, Resources.

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Conflict of interest

The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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Supplementary material

The Supplementary Material for this article can be found online at: https://www.frontiersin.org/articles/10.3389/fnano.2025.1599264/full#supplementary-material

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